



Electroreduction of low concentration CO₂ at atomically dispersed Ni-N-C catalysts with nanoconfined ionic liquids

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ARTICLE INFO

Keywords:
CO₂ electroreduction
Ni-N-C
Ionic liquids
Single atom catalyst
CO₂ capture

ABSTRACT

Most studies on electrochemical CO₂ reduction reaction use pure CO₂, which requires extra energy for CO₂ capture and enrichment from atmosphere. Herein, nanoconfined ionic liquids are introduced into porous atomically dispersed nickel-nitrogen-carbon (Ni-N-C) catalysts to enrich local CO₂ concentration and increase the CO₂RR kinetics. A series of high-CO₂-solubility ionic liquids (ILs) were impregnated into the pores of the columnar Ni-N-C catalyst to alter the CO₂-Ni sites interactions and create a solid/liquid interface with high CO₂ concentration. The optimal Ni-N-C/[Bmim][PF₆] composite outperforms the Ni-N-C catalyst for pure CO₂ electroreduction with a maximum FE_{CO} of 99.6% and 2.7-fold larger j_{CO} . The high solubility of CO₂ in ILs compared to aqueous electrolyte enables direct electrolysis of CO₂ at low concentrations. When fed with 5–10% (v/v) CO₂, the Ni-N-C/[Bmim][PF₆] composite exhibited up to 1.5-fold higher FE_{CO} and a 68% increase of j_{CO} , in comparison to Ni-N-C, and robust stability over 30 h.

1. Introduction

Electrochemical CO₂ reduction reaction (CO₂RR) is a promising technology to reduce CO₂ emissions by converting CO₂ into valuable chemicals, thus mitigating society's dependence on fossil fuels [1,2]. However, CO₂RR is suffering from stable C=O bonds (806 kJ/mol) in linear CO₂ molecules [3], and the large reorganizational energy (−1.9 V vs. standard hydrogen electrode) required for transforming linear CO₂ into bent CO₂[−] during the first single electron transfer, which imposes slow reaction kinetics and high overpotential for CO₂RR [4]. Other challenges include low CO₂ solubility in aqueous electrolytes [2] and strong competition from hydrogen evolution reaction (HER) [5]. The development of efficient electrocatalysts holds the key to address these challenges [6–14]. Single-atom catalysts (SACs) with homogeneous distribution of well-defined metal active sites on the supports have shown great promises for CO₂RR due to their strong atoms-supports interactions, maximum metal utilization and excellent catalytic activity [15–19]. For example, SACs with metal-nitrogen moieties (M-N_x) such as Ni-N SACs have been highly active and selective to CO formation [20,21]. Nonetheless, SACs are generally limited by their single active sites and simple structures, which make it challenging to catalyze complicated reactions involving multiple intermediates such as CO₂RR.

New strategies are needed to functionalize SACs to achieve multi-active sites and/or multi-functionalities.

Up to date, most CO₂RR studies have been using pure CO₂ gas to achieve high electrochemical performances [22–24]. However, industrial CO₂ sources are typically at low concentrations such as that from the fired power plants (5–10% v/v) [25]. Therefore, CO₂ needs to be captured, enriched, and purified to be applied for CO₂RR, which takes a lot of energy and additional cost [26,27]. The direct reduction of low concentration CO₂ is therefore more favorable from an economic prospective [28–30]. ILs have been reported as attractive media for CO₂ capture and electrochemical reduction, attributed to their high CO₂ solubility [31–33], enhanced CO₂ transfer [34], and improved selectivity [35]. Recently, ILs electrolytes have been used for CO₂-to-CO conversion for diluted CO₂ feed, which was demonstrated to be more profitable and economically attractive than using pure CO₂ [27]. However, using ILs electrolytes directly in CO₂ electrolyzers generally suffer from several intrinsic limitations of ILs including high viscosity and poor ionic conductivity [36–38], which subsequently lead to sluggish mass transfer and high energy consumption. Additionally, ionic liquids are relatively costly. Therefore, using large quantity of viscous ILs in CO₂ electrolyzers for CO₂ capture and reduction is technologically and economically impractical.

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Herein, we show an atomically dispersed Ni-N-C catalyst with nanoconfined ILs for efficient CO₂RR in both pure and low concentration CO₂. The nanoconfined ILs play multiple roles as a CO₂ concentrator, a co-catalyst, and a HER suppressor. Using ILs confined in nanoscale pores also solves the high viscosity and low conductivity issues that bulk IL electrolytes have for CO₂RR [36–38]. A series of ILs with different combinations of cations and anions were employed and their effects on CO₂RR are studied. The optimized Ni-N-C/[Bmim][PF₆] composite exhibited promising catalytic performances for electroreduction of CO₂ at low concentration, demonstrating great potentials of the Ni-N-C/ILs composites for future industrial CO₂ electroreduction applications.

2. Experimental section

2.1. Chemicals

Nickel(II) nitrate was purchased from Chem-Supply Pty. Ltd. Melamine was purchased from Sigma-Aldrich Pty Ltd. 1-butyl-3-methylimidazolium tetrafluoroborate ([Bmim][BF₄]), 1-ethyl-3-methylimidazolium triflate ([Emim][CF₃SO₃]), 1-ethyl-3-methylimidazolium tetrafluoroborate ([Emim][BF₄]), 1-ethyl-3-methylimidazolium bis(trifluoromethylsulfonyl)imide ([Emim][TF₂N]), and 1-butyl-3-methylpyridinium bis(trifluoromethylsulfonyl)imide ([BMPy][TF₂N]) were purchased from IoLiTec Ionic Liquids Technologies. 1-butyl-3-methylimidazolium hexafluorophosphate ([Bmim][PF₆]), and methanol were purchased from Chem-Supply Pty. Ltd. 1-butyl-3-methylimidazolium trifluoromethanesulfonate ([Bmim][CF₃SO₃]) was purchased from Sigma-Aldrich Pty Ltd. All reagents and solvents were of analytical grade and used as received without additional purification. The CO₂ and Ar feed gases were purchased from Air Liquide in Australia.

2.2. Preparation of Ni-N-C

In a typical procedure, Nickel(II) nitrate and melamine were mixed together with a mass ratio of 1:10, other mass ratios don't exhibit significant difference in CO₂RR performances (Fig. S1), then annealed at high temperature (1000 °C) under Ar flow. After which, the obtained black powders were washed with HCl, centrifuged, washed with distilled water several times and dried in vacuum at 60 °C overnight, obtaining the columnar Ni-N-C. During the pyrolysis of Ni²⁺ and melamine, melamine was gradually converted to g-C₃N₄ via poly-condensation at 370–600 °C [23]. Meanwhile, the Ni atoms were immobilized onto the defect sites of g-C₃N₄. With increasing the temperature over 700 °C, the carbon growth was catalyzed by the immobilized Ni atoms. The in situ formed g-C₃N₄ as the template guided the formation of atomically dispersed Ni-N-C catalyst.

2.3. Preparation of Ni-N-C/ILs composite catalysts

To prepare the Ni-N-C/ILs composite catalysts, ILs were confined in Ni-N-C catalyst using a simple ultrasonication-centrifugation method. The Ni-N-C catalyst powder (80 mg) was homogeneously dispersed in 4 mL ILs ([Bmim][BF₄], [Bmim][PF₆], [Emim][TF₂N], [BMPy][TF₂N], [Bmim][CF₃SO₃], [Emim][CF₃SO₃], [Emim][BF₄]) under ultrasonication for 1 h at room temperature. Subsequently, the slurry was centrifuged, washed with methanol once to remove the ILs on the outer surface of the Ni-N-C catalyst, and then dried in vacuum at 70 °C overnight, obtaining the Ni-N-C/ILs. The Ni-N-C/ILs composite catalysts prepared from various ILs were denoted as Ni-N-C/[Bmim][BF₄], Ni-N-C/[Bmim][PF₆], Ni-N-C/[Emim][TF₂N], Ni-N-C/[BMPy][TF₂N], Ni-N-C/[Bmim][CF₃SO₃], Ni-N-C/[Emim][CF₃SO₃], and Ni-N-C/[Emim][BF₄].

2.4. Catalyst characterizations

Scanning electron microscope (SEM) images were collected using QUANTA 450. Transmission electron microscopy (TEM), high angle annular dark-field scanning TEM (HAADF-STEM), and energy-dispersive X-ray spectroscopy (EDS) mapping images were obtained by JEOL JEMARM200f microscope at 200 kV. X-ray spectroscopy (XPS) results were obtained by Thermo ESCALAB250i. X-ray diffraction (XRD) was carried out on a PANalytical X'Pert X-ray diffraction system (45 kV, 40 mA, Cu K α radiation). Brunauer-Emmett-Teller (BET) surface areas were measured using Tristar II 3020 instrument by nitrogen adsorption at 77 K. Fourier transform infrared (FT-IR) spectra was collected by Spectrum 100/Spotlight 400. Nuclear Magnetic Resonance (NMR) was conducted using Bruker Advance III 400 (Rabi).

2.5. Electrochemical measurements

CO₂RR measurements were conducted in a H-cell which consisted of a reference electrode (saturated calomel electrode, SCE), a Pt wire counter electrode and a working electrode separated by Nafion membrane. 10 mg catalysts were dispersed into the solution of H₂O (100 μ L), ethanol (300 μ L), and 5 wt% Nafion solution (100 μ L), which were sonicated for 2 h at room temperature. Then, the ink was dropped onto carbon fiber paper and dried in air, obtaining the working electrode with 1 mg/cm² catalyst loading. The electrochemical data were collected in 0.5 M KHCO₃. The potentials were calculated by the Nernst equation ($E_{RHE} = E_{SCE} + 0.0591 \times pH + 0.241$ V, 25 °C).

Chronoamperometry was employed to measure the products and Faradic efficiency of CO₂RR at fixed potentials. The gaseous products were quantified using a gas chromatograph (GC) equipped with a flame ionization detector (FID) for CO and CH₄ and a thermal conductivity detector (TCD) for H₂ quantification. The carrier gas was ultrapure Ar (99.999%). The CO₂ flow rate was 20 sccm controlled by a Cole-Parmer mass flow meter for electroreduction of pure CO₂. While for diluted CO₂ reduction, flow rate ratio of CO₂/(Ar+CO₂) was adjusted to give 5%, 10%, 25%, and 50% CO₂ concentration. GC was calibrated using standard gas mixtures under standard conditions (1 atm, 298 K, Fig. S2).

According to the definition of faradaic efficiency (FE):

$$FE = \frac{\alpha \times F \times n}{Q} = \frac{\alpha \times F \times n}{I_{total} \times t}$$

And ideal gas law:

$$P \times V = n \times R \times T \rightarrow n = \frac{P_o \times V_{volume} \times r}{R \times T_0}$$

Therefore, the FE can be written as:

$$FE = \frac{\alpha \times F}{I_{total}} \times \frac{P_o \times r}{R \times T_0} \times \frac{V_{volume}}{t} = \frac{\alpha \times F \times P_o \times r \times V_{flow_rate}}{R \times T_0 \times I_{total}}$$

Where, α : the number of electrons transferred during reaction ($\alpha_{CO} = \alpha_{H2} = 2$); F: Faradaic constant (96,485 C/mol); P_o: atmospheric pressure (101,325 N/m²); r: gas concentration (ppm), r_{CO} = 0.000110118A_{CO}, r_{H2} = 0.00312A_{H2}; V_{flow} rate: CO₂ volumetric flow rate (sccm); R: ideal gas constant (8.314 N m/(mol K)); T₀: reaction temperature (298.15 K); I_{total}: total current (A). Therefore, FE_{CO} and FE_{H2} were calculated as the following:

$$FE_{CO} = \frac{0.1315 \times V_{flow_rate} \times (0.000110118A_{CO})}{I_{total}} \times 100\%$$

$$FE_{H2} = \frac{0.1315 \times V_{flow_rate} \times (0.00312A_{H2})}{I_{total}} \times 100\%$$

Where, A_{CO}: integral area of CO peak tested by GC; A_{H2}: integral area of H₂ peak tested by GC.

2.6. Electrochemical active surface areas (ECSA) measurements

The ECSA is proportional to double-layer capacitance (C_{dl}) values. C_{dl} was determined in H-type cell by measuring the capacitive current associated with double-layer charging from the scan-rate dependence of cyclic voltammetry (CV). The CV ranged from -0.5 to -0.6 V vs SCE. The scan rates were 10, 20, 40, 60, 80 and 100 mV/s. C_{dl} was estimated by plotting the anodic and cathodic current density difference at -0.55 V vs SCE against the scan rates.

ECSA was calculated based on the definition:

$$\text{ECSA} = C_{dl}/C_s$$

Where C_{dl} corresponds to the slope of the double-layer charging current vs the scan rate plot, and the value of C_s we used is $40 \mu\text{F}/\text{cm}^2$.

3. Results and discussions

3.1. Preparation and characterization of Ni-N-C/ILs composite catalysts

The Ni-N-C catalyst with nanoconfined ILs were fabricated using a

facile impregnation method (Fig. 1a). Firstly, Ni-N-C catalyst was prepared by calcinating a solid mixture of Ni^{2+} and melamine at high temperature to yield a porous columnar carbon structure with uniformly distributed Ni atoms. The Ni-N-C catalyst powder was then dispersed in ILs under ultrasonication at room temperature. ILs were easily confined into the pores of Ni-N-C catalyst by capillary force. The sample was then centrifuged, washed with methanol to remove the ILs adsorbed on the outer surface of the Ni-N-C catalyst, and then dried in vacuum to obtain the Ni-N-C/ILs composite. To systematically investigate the effects of nanoconfined ILs for CO_2 RR, we deployed a series of ILs with different combinations of cations and anions, i.e., $[\text{Bmim}][\text{BF}_4]$, $[\text{Bmim}][\text{PF}_6]$, $[\text{Emim}][\text{TF}_2\text{N}]$, $[\text{BMPy}][\text{TF}_2\text{N}]$, $[\text{Bmim}][\text{CF}_3\text{SO}_3]$, $[\text{Emim}][\text{CF}_3\text{SO}_3]$, $[\text{Emim}][\text{BF}_4]$ (Fig. S3, Table S1).

SEM (Fig. 1b) and TEM (Fig. 1d) images revealed that the Ni-N-C/ $[\text{Bmim}][\text{PF}_6]$ composite presented a columnar morphology with abundant pores, and exhibited isolated Ni atoms on carbon support, shown as the bright spots in the HAADF-STEM picture (Fig. 1c). Similar results were observed for the Ni-N-C catalyst (Fig. S4a–b). Ni nanoparticles were also detected, which appear as the dark parts in the HAADF-STEM (Figs. S4c, S5). Moreover, the energy-dispersive X-ray spectroscopy

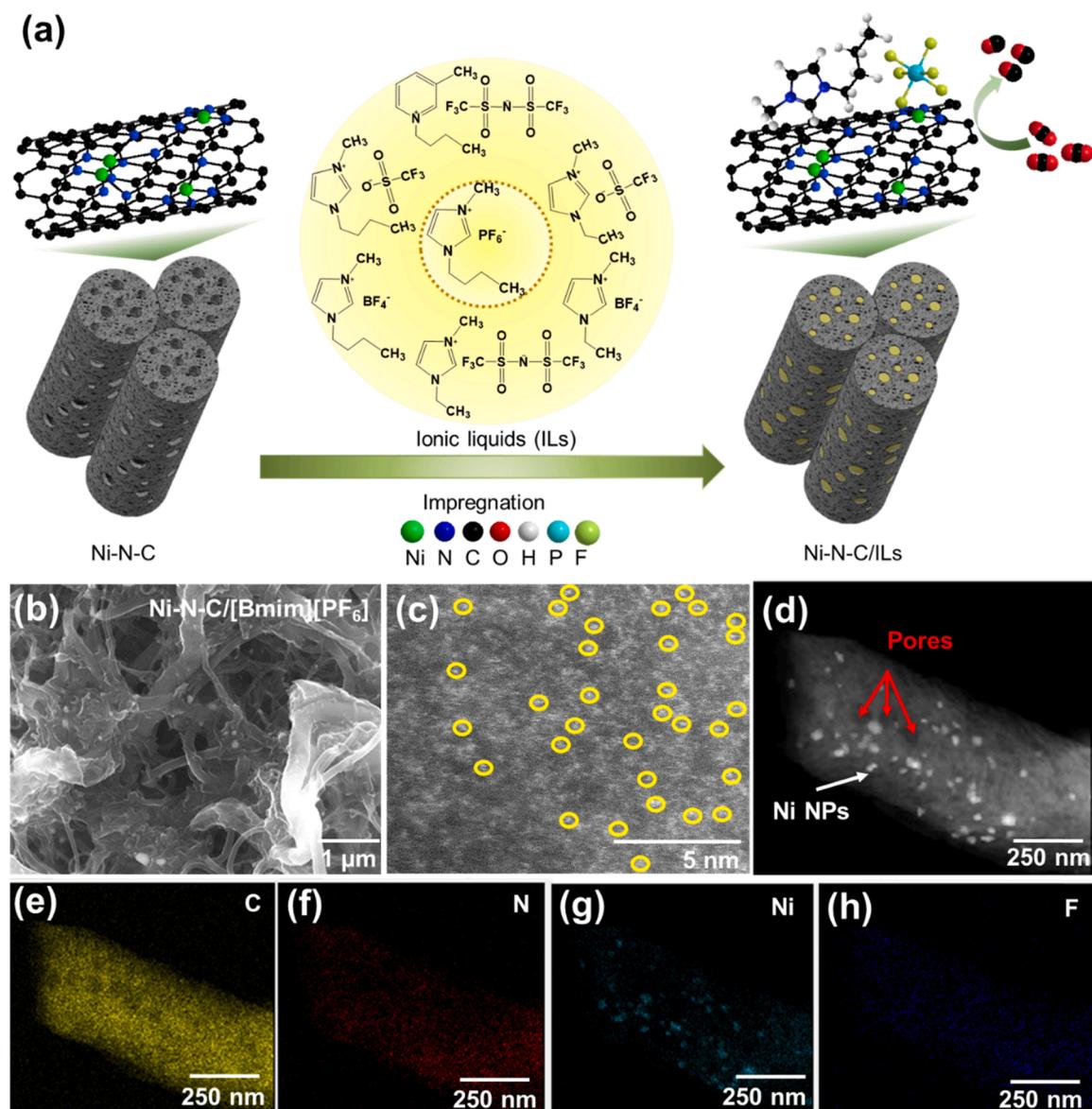


Fig. 1. Synthesis and structural characterization. (a) Preparation of Ni-N-C/[Bmim][PF₆] composite catalysts. (b) SEM of Ni-N-C/[Bmim][PF₆]. (c) HAADF-STEM of Ni single atoms in Ni-N-C/[Bmim][PF₆]. (e-h) EDS mapping of Ni-N-C/[Bmim][PF₆] for (d).

(EDS) suggested homogeneous dispersion of C, N, Ni from the Ni-N-C host and F from the ILs (Figs. 1e-h, S4d-g).

As shown in the X-ray diffraction (XRD) spectra, the Ni-N-C catalyst and Ni-N-C/ILs composites exhibited similar carbon peaks at 26.5°, 44.6°, 74.2° for (002), (101), (112) planes, and two nickel peaks for (111) and (200) planes attributed to the Ni nanoparticles (Fig. 2a, S6). X-ray photoelectron spectroscopy (XPS) was performed to analyze the element state of catalyst surface and composition. The Ni 2p XPS spectrum of Ni-N-C and Ni-N-C/[Bmim][PF₆] was dominated by the peak at 854.8 eV (Fig. S7a), which was lower than the Ni²⁺ (856 eV) in nickel phthalocyanine (NiPc) [39], assigning to the low-valent Ni^{δ+} in the catalysts. Besides, the N 1s spectra indicated the presence of pyridinic N (398.92 eV), Ni-N (400.05 eV), pyrrolic N (401.25 eV) and graphitic N (404.53 eV) in the Ni-N-C and Ni-N-C/[Bmim][PF₆] samples (Fig. S7b). Fourier transform infrared (FT-IR) spectra for the Ni-N-C/[Bmim][PF₆] sample shows the characteristic vibration of methine (C-H) in [Bmim]⁺ (3172, 1386, 1169, 748 cm⁻¹), stretching vibration of [PF₆]⁻ (842 cm⁻¹) and the C-N vibration inside [Bmim][PF₆] (1574 cm⁻¹) [40, 41], suggesting the successful introduction of ILs into the Ni-N-C catalyst (Figs. 2c, S8).

Brunauer-Emmett-Teller measurements were performed to investigate the specific surface area and pore size distribution of the samples. The Ni-N-C catalyst showed a surface area of 123 m²/g with mesopores at 2.3–21.6 nm (Fig. 2b, S9). The Ni-N-C/[Bmim][PF₆] composite exhibited lower surface area (66 m²/g) and mesopores (2.5–11.0 nm), indicating the successful impregnation of [Bmim][PF₆] into the interior pores of the Ni-N-C host. To estimate the ILs content in the composite,

the masses of the catalysts before and after introducing nanoconfined ILs were denoted as m₁ and m₂, respectively. The estimated ILs contents were calculated by the ratio of (m₂-m₁)/m₂. The Ni-N-C/[Bmim][PF₆] composite delivered a relatively large IL content (45.58 wt%, Table S2), favorable to improving the CO₂ concentration in aqueous electrolyte. Furthermore, electrochemical active surface area (ECSA), estimated from the double-layer capacitance (Figs. S10–S11a) [42], also showed that the Ni-N-C/[Bmim][PF₆] composite has a lower ECSA value than the Ni-N-C catalyst (Fig. S11b) due to the blocked pores in the Ni-N-C by [Bmim][PF₆] impregnation.

3.2. Electroreduction of pure CO₂ using Ni-N-C/ILs composite catalysts

CO₂RR measurements were carried out in a three-electrode H-cell using CO₂-saturated 0.5 M KHCO₃ as the electrolyte (Fig. S12). In this work, all samples were washed by 3 M HCl to remove the Ni nanoparticles outside the carbon nanotube. The remained Ni nanoparticles were wrapped inside the carbon nanotube which do not directly participate in the CO₂ electroreduction [43,44]. The excellent performances of Ni-N-C/[Bmim][PF₆] on CO₂ electroreduction was therefore attributed to atomically dispersed Ni sites (Figs. 3 a, b, 4). To further confirm this and elucidate the role of Ni nanoparticles (NPs), the control experiment was performed. Two Ni-N-C samples were prepared with and without acid washing, which were denoted as Ni-N-C (acid washing) and Ni-N-C (no acid washing), respectively. According to the SEM images (Fig. S13a-b), Ni-N-C (acid washing) presented clear carbon nanotube structure and few Ni NPs on the surface of carbon nanotubes,

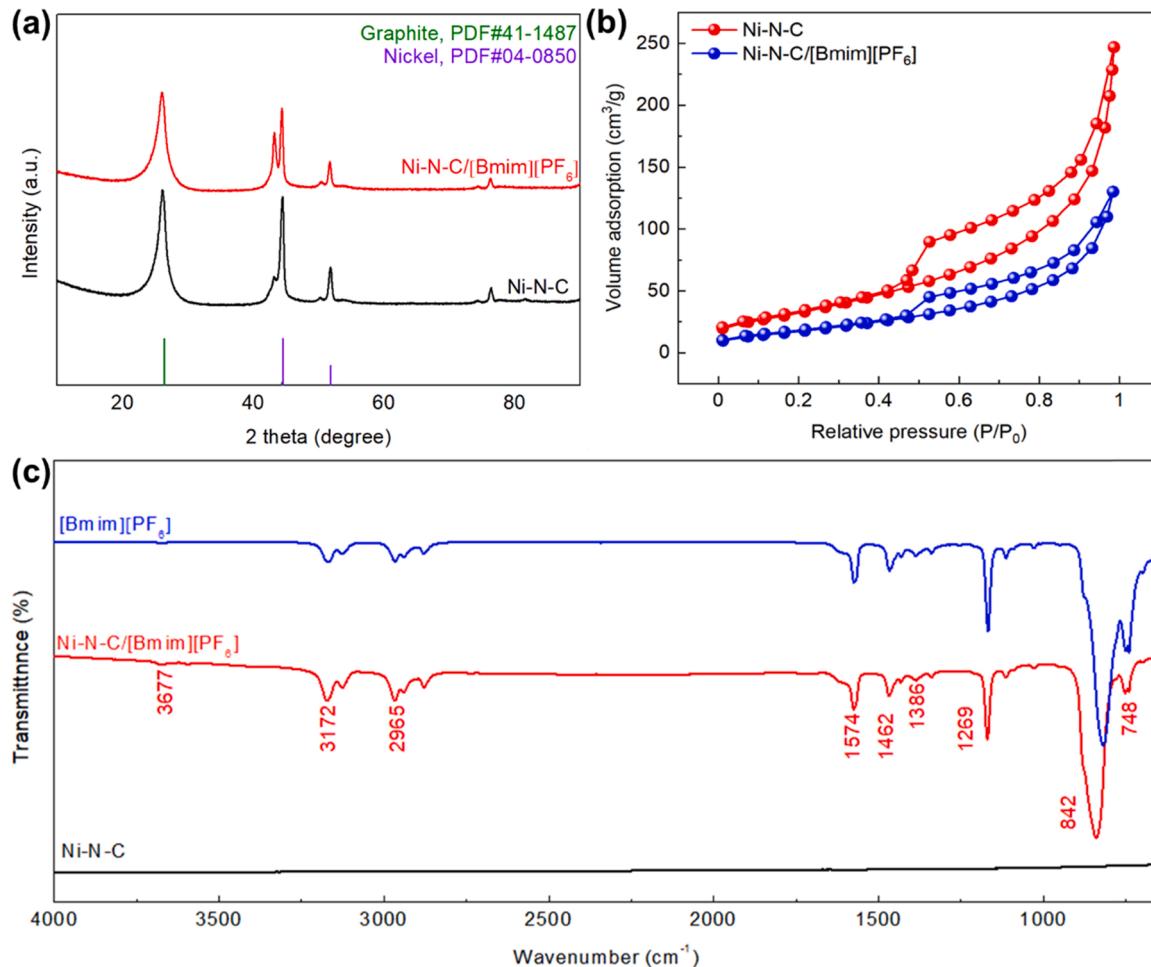


Fig. 2. (a) XRD of Ni-N-C catalyst and Ni-N-C/[Bmim][PF₆] composite catalysts. (b) Nitrogen adsorption-desorption isotherms for Ni-N-C, and Ni-N-C/[Bmim][PF₆]. (c) IR of Ni-N-C, Ni-N-C/[Bmim][PF₆], and pure [Bmim][PF₆].

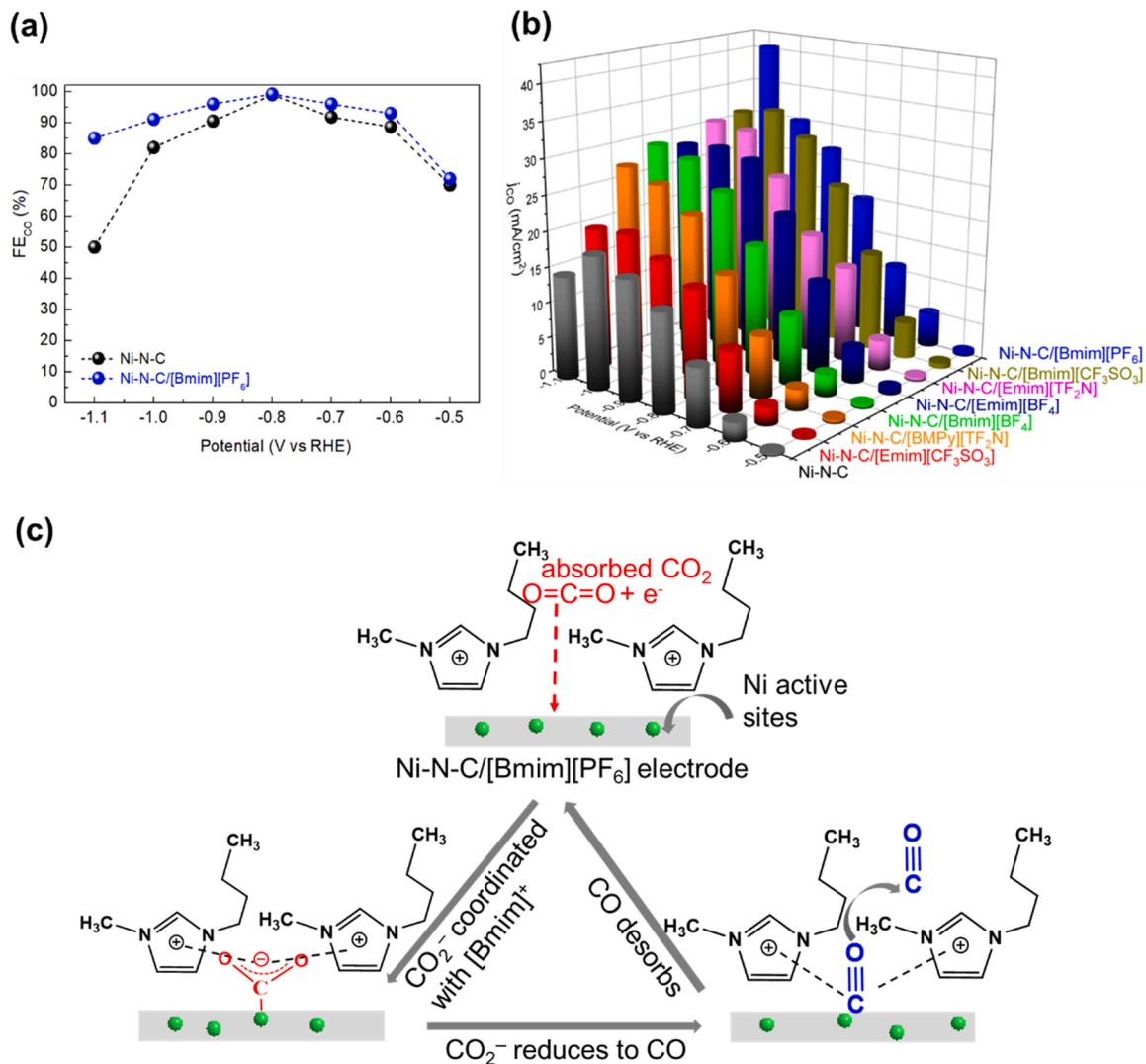


Fig. 3. (a) FE_{CO} of Ni-N-C catalyst and Ni-N-C/[Bmim][PF₆]. (b) *j*_{CO} of Ni-N-C catalyst and Ni-N-C/ILs composite catalysts. (c) Mechanism of CO₂RR on Ni-N-C/[Bmim][PF₆] electrodes.

while Ni-N-C (no acid washing) exhibited abundant Ni NPs on the surface. XRD results revealed stronger Ni NPs peaks of Ni-N-C (no acid washing) than that of Ni-N-C (acid washing), the larger intensity ratio of Ni NPs peak to carbon peak (Ni(111)/C(002)) for Ni-N-C (no acid washing) verified more Ni NPs on carbon nanotube (Fig. S13c). However, the Ni-N-C (acid washing) and Ni-N-C (no acid washing) presented similar CO₂RR performance (Fig. S14), which demonstrated that Ni NPs do not contribute significantly to CO₂RR. This is consistent with previous studies that Ni NPs don't present significant activity for CO₂RR [43, 44].

It is found that all Ni-N-C/ILs composites outperform the Ni-N-C catalyst over a wide potential range from -0.5 V to -1.1 V vs RHE, delivering higher FE_{CO} (Figs. 3a, S15a), lower Faradaic efficiency of H₂ (FE_{H2}, Fig. S15b), and larger CO partial current density (*j*_{CO} per geometric area, Fig. 3b). Among various Ni-N-C/ILs composites, Ni-N-C/[Bmim][PF₆] reached a maximum FE_{CO} of 99.6% at -0.8 V vs RHE, slightly higher than that of Ni-N-C catalyst (98.8%). Notably, at more negative potential of -1.1 V vs RHE, it also exhibited high FE_{CO} of 85.0% and maximum *j*_{CO} of 40.0 mA/cm², which were 1.7- and 2.7-fold higher than that of Ni-N-C at same applied potential, respectively, which meant higher selectivity to CO than HER by the nanoconfined [Bmim][PF₆]. This indicated the effects of the nanoconfined [Bmim][PF₆] in suppressing H₂ and offering higher selectivity to CO at high applied

potential. Besides, the nanoconfined [Bmim][PF₆] into Ni-N-C catalyst also contributed to better CO₂RR performance than some other efficient catalysts using electrolytes of ILs as summarized in Table S4. Additionally, the CO partial current density normalized by ECSA (*j*_{CO} per ECSA, Fig. S15c) for the Ni-N-C/ILs is also larger than that of Ni-N-C catalyst, suggesting the intrinsic activity of each active site of Ni-N-C catalyst is enhanced for CO₂RR upon ILs confinement, rather than a simple surface area effect. The surface area effect normally means the larger surface area, the more active sites, and the higher electrochemical performance. However, in our system the Ni-N-C/[Bmim][PF₆] exhibited a lower surface area but better electrochemical performances than the Ni-N-C catalyst. Therefore, the enhanced electrochemical performances are not purely originated from enlarged surface area but also the ionic liquid co-catalysis effect. Nyquist plots (Fig. S16) were conducted to study the ohmic resistance of system (*R*_s) and mass transfer property of catalysts. In the bulk [Bmim][PF₆] electrolyte, large *R*_s (~211.08 Ω) and diffusion coefficient (slopes at low-frequency) deteriorated significantly because of the high viscosity (310 mPa·s) of ILs [37,45]. While in 0.5 M KHCO₃ aqueous electrolyte, Ni-N-C/ILs and Ni-N-C showed small *R*_s (1.77–10.31 Ω) at high-frequency region (Table S3) and similar slopes at low-frequency region, demonstrating the ILs confinement has negligible effect on the electronic conductivity and the system diffusivity. This can also be elucidated by the Bode plots (Fig. S17) where two

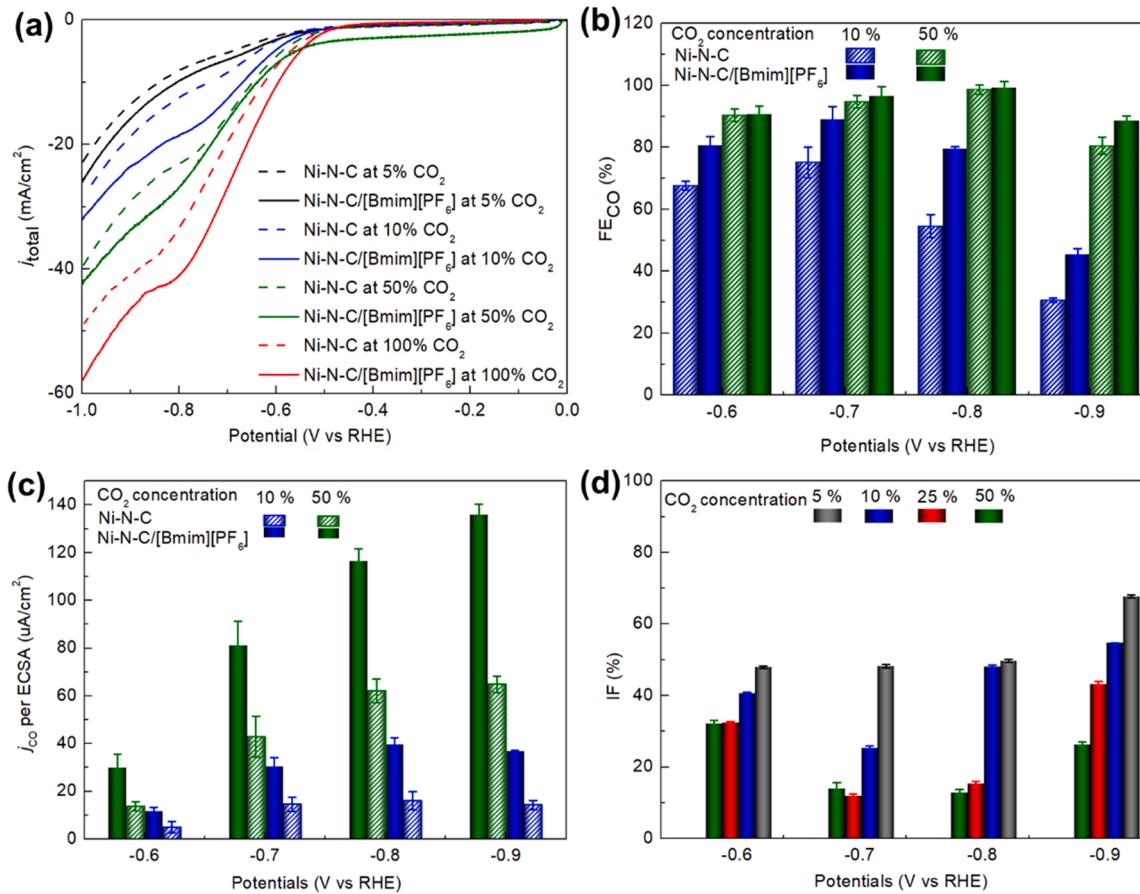


Fig. 4. Electrochemical performances of Ni-N-C/[Bmim][PF₆] for electroreduction of diluted CO₂. (a) LSV curves in diluted CO₂ (5%, 10%, 50%) and pure CO₂. (b) FE_{CO} of Ni-N-C and Ni-N-C/[Bmim][PF₆] in 10% and 50% CO₂. (c) *j*_{CO} per ECSA of Ni-N-C and Ni-N-C/[Bmim][PF₆] in 10% and 50% CO₂. (d) IF values for Ni-N-C/[Bmim][PF₆] based on Ni-N-C in 5%, 10%, 25%, and 50% CO₂ concentrations at different applied potentials. Error bars indicate the standard deviation of three independent samples.

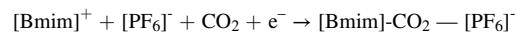
slopes (two time-constants upon decreasing frequency) for the kinetic and diffusion process were observed, respectively. Both Ni-N-C/ILs and Ni-N-C samples displayed similar slopes for the kinetic process at higher frequency, while slightly higher slopes of the diffusion process of the Ni-N-C/ILs were detected. The peaks at lower frequency regions were featured by the charge transfer to Faradic adsorbed CO₂ reduction intermediates on the electrode surface. All these indicated similar charge transfer kinetics of Ni-N-C/ILs and Ni-N-C, signifying the negligible effects of the ILs modifications on the conductivity and diffusivity.

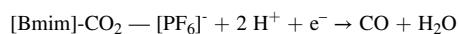
To understand the effects of IL structures on CO₂RR activity, *j*_{CO} values of the Ni-N-C/ILs composites with different IL anions or cations were compared. It was found the *j*_{CO} values obtained generally follow the trend of CO₂ solubilities in different ILs. For example, using the ILs with same anions such as [CF₃SO₃]⁻ and [BF₄]⁻, the *j*_{CO} values follow the order of Ni-N-C/[Bmim][CF₃SO₃] > Ni-N-C/[Emim][CF₃SO₃] > Ni-N-C (Fig. S18) and Ni-N-C/[Bmim][BF₄] > Ni-N-C/[Emim][BF₄] > Ni-N-C (Fig. S19). The longer chain lengths of the imidazolium cations (e.g. [Bmim]⁺ vs [Emim]⁺) is reported to result in the higher CO₂ solubility [46]. For the ILs with same cations such as [Bmim]⁺ and [Emim]⁺, the following *j*_{CO} orders of Ni-N-C/[Bmim][PF₆] > Ni-N-C/[Bmim][BF₄] > Ni-N-C (Fig. S20) and Ni-N-C/[Emim][Tf₂N] > Ni-N-C/[Emim][BF₄] > Ni-N-C have been found (Fig. S21), which also broadly follow the CO₂ solubility trends in the ILs (Table S1) [38,47].

However, CO₂ solubility is found not the only factor that determines the catalytic activity of the Ni-N-C/ILs. Other factors contribute as well. For example, the CO₂ solubility in [Bmim][PF₆] and [Emim][Tf₂N] is lower than that in [BMpy][Tf₂N] (Table S1). However, the Ni-N-C/[Bmim][PF₆] and Ni-N-C/[Emim][Tf₂N] composites exhibited higher

activity and *j*_{CO} than Ni-N-C/[BMpy][Tf₂N] (Fig. S22). In this case, the co-catalyst effect of the nanoconfined ILs is believed to play an important role. It has been established that imidazolium-based ILs e.g. [Emim]⁺ and [Bmim]⁺ can promote CO₂RR by stabilizing the intermediate CO₂^{*} via the formation of [Emim]-CO₂ and [Bmim]-CO₂ complex during the CO₂RR process (Fig. 3c), increasing the local CO₂ concentration near cathode surface, and lowering the energy barrier for CO₂ reduction [48,49].

Fig. 3c illustrated the plausible mechanism for the enhancement of electrochemical performance CO₂RR by the introduction of the nanoconfined ILs. Normally, during the CO₂RR process catalyzed by atomically distributed Ni-N-C catalyst, the surface absorbed CO₂ was transformed into COOH* intermediate via a concerted H⁺/e⁻ transfer and further reduced to CO via a second H⁺/e⁻ transfer, or the *COOH intermediate was generated via a decoupled H⁺/e⁻ transfer with the formation of the surface absorbed CO₂^{*} radical which were the typical CO₂RR pathways for CO formation (Fig. S23) [50,51]. With introducing the nanoconfined ILs such as [Bmim][PF₆], the [Bmim]⁺ cations can coordinate with the CO₂^{*} radicals to form [Bmim]-CO₂ complexes, providing a low-energy pathway for CO production [48,49]. This is different from Ni-N-C catalyst where the rate-limiting step of one electron transfer to CO₂ form a high-energy intermediate [49]. The [Bmim]-CO₂ complex was finally converted to adsorbed CO followed by CO desorption (Fig. 3c). Therefore, the cathodic reaction by Ni-N-C/[Bmim][PF₆] was different from that without ILs (2CO₂ + 2H⁺ + 2e⁻ → CO + H₂O) [48,49,52]. With ILs such as [Bmim][PF₆] [49]:





Most ILs, if not all, including both hydrophobic and hydrophilic ILs, have some degree of water miscibility [53]. Therefore, the ILs cannot block the contact of water or ions to the catalyst. Here in our case, even the hydrophobic $[\text{Bmim}]\text{[PF}_6]$ still can have a water miscibility up to 2.67 ± 0.03 wt% [54]. Water present in $[\text{Bmim}]\text{[PF}_6]$ is in contact with the catalyst surface and act as a proton source. This has been established in previous studies where $[\text{Bmim}]\text{[PF}_6]$ is directly as bulk electrolytes for CO_2RR [46,55]. $[\text{Bmim}]\text{[BF}_4]$, $[\text{Emim}]\text{[BF}_4]$, $[\text{Bmim}]\text{[CF}_3\text{SO}_3]$, and $[\text{Emim}]\text{[CF}_3\text{SO}_3]$ are hydrophilic, $[\text{Bmim}]\text{[PF}_6]$, $[\text{BMPy}]\text{[TF}_2\text{N}]$, and $[\text{Emim}]\text{[TF}_2\text{N}]$ are hydrophobic ILs. In this work, as hydrophilic ILs can dissolve in aqueous solutions [56,57], hydrophobic ILs are generally preferred as they are more stable in aqueous 0.5 M KHCO_3 electrolyte and stably confined into the porous Ni-N-C catalyst, thereby offering a steady electrolysis performance. The hydrophobicity of ILs can be further tuned by altering the combinations of anions and cations. For example, the longer alkyl chain length of the cations, the stronger hydrophobicity. The stronger the hydrogen bonding anions could form (anion...HOH...anion) with water (e.g. $[\text{PF}_6]^- < [\text{TF}_2\text{N}]^- < [\text{BF}_4]^- < [\text{CF}_3\text{SO}_3]^-$), the more hydrophobic the ionic liquids are [53,58]. In our study, among the ionic liquids applied, the $[\text{Bmim}]\text{[PF}_6]$ composite exhibited the best electrochemical performance for CO_2 reduction (Fig. 3b). The nanoconfined $[\text{Bmim}]\text{[PF}_6]$ is found quite stable and do not have significant loss during CO_2RR measurement, as confirmed by ^1H NMR and ^{13}C NMR (Fig. S24).

3.3. Electroreduction of diluted CO_2

The high solubility of CO_2 in ILs have made them attractive media for CO_2 capture [31–33], which inspired us to explore CO_2 capture and electroreduction at the same Ni-N-C/IL composite catalysts. Thus, diluted CO_2 of 5%, 10%, 25% and 50% (v/v) were prepared by mixing CO_2 with Ar at different flow rate ratios and applied to the electrochemical cell. Ni nanoparticles were not the main factor contributing to the CO_2RR under low CO_2 partial pressure, which was confirmed by the similar electrochemical performances of diluted CO_2 reduction for the Ni-N-C catalysts with and without acid washing (Fig. S25). Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ composite was selected for electroreduction of low concentration CO_2 due to its high activity and selectivity (e.g. j_{CO} and FE_{CO}), as a result of its high CO_2 solubility and the co-catalyst effect. As shown in linear sweep voltammetry (LSV, Fig. 4a), the total current density (j_{total}) for both Ni-N-C catalyst and Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ composite increase as expected upon increasing CO_2 concentration from 5% to 100%. In comparison to Ni-N-C, the Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ catalyst always shows larger j_{total} (Fig. 4a), higher FE_{CO} (Figs. 4b, S26), larger j_{CO} per geometric area (Fig. S27) and j_{CO} per ECSA (Figs. 4c, Fig. S28) than the Ni-N-C catalyst, as expected, attributed to the high CO_2 solubility in $[\text{Bmim}]\text{[PF}_6]$, the co-catalytic effect, and CO_2 preconcentration by $[\text{Bmim}]\text{[PF}_6]$. For example, at low CO_2 concentrations of 10%, Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ achieved maximum FE_{CO} of 89.1% with j_{CO} of 4.91 mA/cm^2 at -0.7 V vs RHE, which is comparable to other reported effective catalysts for diluted CO_2RR (Table S5).

To further understand the preconcentration effects of $[\text{Bmim}]\text{[PF}_6]$ on j_{CO} , the increase factor (IF) of the CO partial current density against Ni-N-C catalyst ($\text{IF} = ([j_{\text{CO}(\text{Ni-N-C/Bmim})}\text{[PF}_6] - j_{\text{CO}(\text{Ni-N-C})})/j_{\text{CO}(\text{Ni-N-C})} \times 100\%$) was determined (Fig. 4d). It was found that larger IF values were always obtained for Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ at lower CO_2 concentrations. For example, when using 50% (v/v) CO_2 , IF values of 26%, 13%, 14% and 32% are achieved at -0.9 , -0.8 , -0.7 and -0.6 V vs RHE, respectively. While at 5% (v/v) CO_2 , the j_{CO} with maximum IF values of 68%, 50%, 48%, and 48% were obtained at -0.9 , -0.8 , -0.7 and -0.6 V vs RHE, respectively. The higher increase factor obtained at lower CO_2 concentration can be attributed to the high solubility and local pre-concentration of CO_2 in $[\text{Bmim}]\text{[PF}_6]$. Additionally, the IF values at different CO_2 concentrations exhibited a decreasing trend with

the potential changing from -0.6 V to -0.7 V vs RHE, then followed by a gradual increase upon more negative potentials from -0.7 V vs -0.9 V vs RHE. This behavior is related to the intrinsic activity of the atomically dispersed Ni-N-C catalyst for CO_2RR which are more active around -0.7 and -0.8 V vs RHE [52,59]. Moreover, stability testing was performed at -0.8 V vs RHE with 10% (v/v) CO_2 , a typical CO_2 concentration in the waste gas from fire power plants. The Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ presented stable FE_{CO} and j_{CO} over 30 h continuous CO_2 electrolysis (Fig. S29). Slight difference between the FE_{CO} by stability testing (Fig. S29) and that in 10% CO_2 concentration (Fig. 4b) was likely due to the difference of catalysts form batch to batch, as well as the impregnation process for confinement of ionic liquid.

4. Conclusions

We developed a facile nanoconfined ILs strategy for improving the reduction of low concentration CO_2 . Among a series of Ni-N-C/ILs composites, Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ achieved the best catalytic performance for pure CO_2 reduction with maximum FE_{CO} of 99.6% and j_{CO} of 40 mA/cm^2 at -0.8 V and -1.1 V vs RHE, respectively. When applied for diluted CO_2 reduction, Ni-N-C/ $[\text{Bmim}]\text{[PF}_6]$ exhibited higher FE_{CO} and j_{CO} than the atomically dispersed Ni-N-C catalyst, as well as excellent stability over 30 h under a low concentration of 10%. These excellent performances are attributed to the high CO_2 solubility in ionic liquids, intrinsic activity of single atomic Ni-N-C sites, the large surface area of carbon nanotubes to expose abundant Ni sites, fast electron/mass transfer facilitated by carbon nanotubes, and the co-catalyst effect of the ionic liquid. Because ionic liquids are designable and have been extensively used for CO_2 capture. It is expected that ionic liquids with higher CO_2 solubility and suitable combinations of cations and anions can be designed to further enhance the conversion of low concentrations CO_2 from various industrial sources. The nanoconfined ILs also provide simple yet versatile strategy for tuning the properties of atomically dispersed metal-nitrogen sites for a range of catalytic applications.

CRediT authorship contribution statement

Qian Sun: Conceptualization, Validation, Data curation, Investigation, Formal analysis, Writing – original draft. **Yong Zhao:** Writing – review & editing. **Wen-hao Ren:** Writing – review & editing. **Chuan Zhao:** Writing – review & editing, Supervision, Funding acquisition.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgements

This study was supported by the Australian Research Council (FT170100224, DP210103892, IC200100023). We also thank the UNSW Digital Grid Futures Institute for the award of 2021 Interdisciplinary Seed Fund, and the UNSW Mark Wainwright Analytical Centre (MWAC) for providing access to their XRD, SEM, TEM, XPS and facilities.

Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2021.120963.

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